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using classical solution chemistry (Müller & Sievert, 1974; Müller, Diemann & Heidborn, 1970). The crystal structures of these compounds were reported about 20 years ago. It is interesting to note that these preparation methods did not lead to the successful synthesis of the potassium selenide. During our research on the preparation of new tungsten chalcogenides using the so-called 'reactive flux' method (Sunshine, Kang & Ibers, 1987; Kanatzidis & Sutorik, 1995), the previously unknown ternary tungsten selenide K_2WSe_4 was obtained. It is most likely that conventional preparation methods, either in aqueous solution or at high temperatures, are destined to favour the thermodynamically more stable binary compounds (Kanatzidis, 1990).

The structure of K_2WSe_4 consists of slightly distorted WSe_4^{-} tetrahedra located on crystallographic mirror planes, with W—Se bond lengths ranging from 2.3050 (7) to 2.3266 (9) Å and Se—W—Se angles between 107.16 (4) and 111.79 (4)°. These data are within the range reported for other ternary tungsten tetraselenides (Müller, Krebs & Beyer, 1968). The K⁺ ions are coordinated by eight and nine Se centres with K—Se distances between 3.263 (2) and 3.722 (3) Å (Fig. 1). The coordination sphere of K1 (CN = 9) is of



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K₂WSe₄

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Abstract

The crystal structure of dipotassium tungsten tetraselenide is based on isolated, slightly distorted WSe₄ tetrahedra joined via K⁺ ions. The compound is isotypic with the previously reported Rb and Cs analogues and crystallizes in the β -K₂SO₄ structure type.

Comment

Ternary alkali metal tungsten chalcogenides A_2WQ_4 (A = Rb, Cs, NH⁴; Q = S, Se) have been prepared

Fig. 1. The coordination spheres of K1 and K2 in K_2WSe_4 . Displacement ellipsoids are plotted at the 50% probability level.

K١

K2

W1 Sel

Se2

Se3

irregular shape whereas K2 (CN = 8) is located within a distorted bis-disphenoid.

The K2Se₈ polyhedra share edges with three WSe₄²⁻ tetrahedra and have common corners with two WSe₄²⁻ units. The K1Se₉ polyhedra are connected to three WSe₄²⁻ tetrahedra *via* edges and to three others *via* corners. As expected, the lattice parameters are smaller than those reported for the analogous Rb and Cs compounds (Table 3) (Müller & Sievert, 1974).

Experimental

Single crystals of K_2WSe_4 suitable for X-ray investigation were synthesized by heating a mixture of K_2Se_3 , Se and W (molar ratio 3:6:1, prepared in a glove box) in sealed and evacuated (10^{-4} bar; 1 bar = 10^5 Pa) glass ampoules over a period of 5 d at 623 K. The crystals were grown by slow cooling of the ampoule (3 K h^{-1}). The friable dark red slabs were manually selected from the melt. K_2WSe_4 is soluble in polar solvents like DMF or acetonitrile. The crystals are moisture-sensitive and decompose within approximately 4 d, and were therefore coated with epoxy glue before being mounted for analysis.

Crystal data

K ₂ WSe ₄	Mo $K\alpha$ radiation
$M_r = 577.89$	$\lambda = 0.71073 \text{ Å}$
Orthorhombic	Cell parameters from 94
Pnma	reflections
<i>a</i> = 9.7337 (8) Å	$\theta = 17-24.5^{\circ}$
b = 7.2012 (6) Å	$\mu = 30.414 \text{ mm}^{-1}$
c = 12.5976(12) Å	T = 293 (2) K
$V = 883.02(13) \text{ Å}^3$	Slab
Z = 4	$0.48 \times 0.22 \times 0.05$ mm
$D_x = 4.347 \text{ Mg m}^{-3}$	Dark red
D_m not measured	

Data collection

Stoe AED-II diffractometer	$R_{\rm int} = 0.0362$
ω –2 θ scans	$\theta_{\rm max} = 29.99^{\circ}$
Absorption correction:	$h = 0 \rightarrow 13$
numerical	$k = -10 \rightarrow 1$
$T_{\rm min} = 0.0072, T_{\rm max} =$	$l = -2 \rightarrow 17$
0.2196	4 standard reflections
1752 measured reflections	frequency: 120 min
1381 independent reflections	intensity decay: neglible
-	

 $\Delta \rho_{\rm max} = 2.33 \ {\rm e} \ {\rm \AA}^{-3}$

 $\Delta \rho_{\rm min} = -2.48 \ {\rm e} \ {\rm \AA}^{-3}$

Extinction correction:

Extinction coefficient:

Scattering factors from

1993)

0.0078 (4)

SHELXL93 (Sheldrick,

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Refinement

Refinement on F^2 $R[F^2 > 2\sigma(F^2)] = 0.032$ $wR(F^2) = 0.086$ S = 1.0261381 reflections 41 parameters $w = 1/[\sigma^2(F_o^2) + (0.0527P)^2]$ where $P = (F_o^2 + 2F_c^2)/3$ $(\Delta/\sigma)_{max} = 0.001$

Table 1. Fractional atomic coordinates and equivalent isotropic displacement parameters $(Å^2)$

$$U_{\rm eq} = (1/3) \sum_i \sum_j U^{ij} a_i^* a_j^* \mathbf{a}_i \cdot \mathbf{a}_j.$$

x	v	z	U_{eq}
0.4537 (2)	1/4	0.3422 (2)	0.0399 (5)
0.1574 (2)	1/4	0.6157 (3)	0.0501 (7)
0.74805 (3)	1/4	0.56811(3)	0.02196 (14)
0.51526 (8)	1/4	0.61001 (9)	0.0388 (3)
0.85321 (7)	0.50758 (10)	0.64021 (8)	0.0427 (2)
0.78380(10)	1/4	0.38718 (9)	0.0428 (3)

Table 2. Selected geometric parameters (Å, °)

K1—Se3	3.263 (2)	K2—Se3 ⁱⁱⁱ	3.6460 (5)
K1—Se3 ⁱ	3.329 (3)	K2—Se3 ^{vi}	3.6460 (5)
K1—Se1	3.427 (3)	K2—Se2 ⁱⁱ	3.667 (3)
K1Se2 ⁱⁿ	3.467 (2)	K2Se2 ⁱⁱⁱ	3.667 (3)
K1—Se2 ⁱⁿ	3.467 (2)	K2—Se11x	3.722 (3)
K1—Se2 ^{iv}	3.613 (2)	K2—W1 ^{vm}	4.029 (2)
K1Se2	3.613 (2)	K2—W1 ^{ix}	4.080(3)
K1Se1 ¹¹¹	3.6632 (6)	K2K1*	4.719(3)
K1—Se1 ^{vi}	3.6632 (6)	W1—Sc2	2.3050(7)
K2—Se1	3.484 (2)	W1—Se2 ^{x1}	2.3050(7)
K2—Se2 ^{vu}	3.507 (2)	W1—Se3	2.3056 (12
K2—Se2 ^{vm}	3.507 (2)	W1—Se1	2.3266 (9)
Se2—W1—Se2 ^{xi}	107.16 (4)	Se2—W1—Se1	110.07 (3)
Se2—W1—Se3	108.82 (3)	Se2 ^{x1} —W1—Se1	110.07 (3)
Se2 ^{x1} —W1—Se3	108.82 (3)	Se3—W1—Se1	111.79 (4)

Symmetry codes: (i) $x - \frac{1}{2}$, y, $\frac{1}{2} - z$; (ii) 1 - x, $y - \frac{1}{2}$, 1 - z; (iii) 1 - x, 1 - y, 1 - z; (iv) $\frac{3}{2} - x$, 1 - y, $z - \frac{1}{2}$; (v) $\frac{3}{2} - x$, $y - \frac{1}{2}$, $z - \frac{1}{2}$; (vi) 1 - x, -y, 1 - z; (vii) x - 1, $\frac{1}{2} - y$, z; (viii) x - 1, y, z; (ix) $x - \frac{1}{2}$, y, $\frac{3}{2} - z$; (x) $\frac{1}{2} - x$, 1 - y, $\frac{1}{2} + z$; (xi) x, $\frac{1}{2} - y$, z.

Table 3. Comparison of the lattice parameters (Å) and cell volumes (Å³) of K_2WSe_4 , Rb_2WSe_4 and Cs_2WSe_4

	а	Ь	С	V
K₂WSe₄	9.734(1)	7.201 (1)	12.598 (1)	883.02(13)
Rb ₂ WSe₄	10.10	7.28	12.85	944.7
Cs ₂ WSe ₄	10.32	7.45	13.07	1004.5

The data were corrected for Lorentz and polarization effects. All atoms were refined with anisotropic displacement parameters. The largest peaks in the difference map are located in the neighbourhood of W1. It is noted that most of the atoms in special positions show significantly higher values for U^{22} than for the other displacement components, indicating pos-sible disorder which could not be resolved during this study. Such disorder is well documented for different compounds with the β -K₂SO₄ structure.

Data collection: *DIF*4 (Stoe & Cie, 1992). Cell refinement: *DIF*4. Data reduction: *DIF*4. Program(s) used to solve structure: *SHELXS*86 (Sheldrick, 1990). Program(s) used to refine structure: *SHELXL*93 (Sheldrick, 1993). Molecular graphics: *SHELXTL/PC* (Sheldrick, 1995). Software used to prepare material for publication: *SHELXL*93 *CIFTAB*.

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Lists of structure factors, anisotropic displacement parameters and complete geometry have been deposited with the IUCr (Reference: BR1161). Copies may be obtained through The Managing Editor, International Union of Crystallography, 5 Abbey Square, Chester CH1 2HU, England.

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A New Polymorph of Li₄P₂O₇

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Abstract

The purpose of the present paper is to report the crystal structure of a new phase of tetralithium diphosphate identified by single-crystal X-ray diffraction. The structure consists of layers built from Li–O tetrahedra and pyrophosphate groups connected to each other through edges and vertices.

Comment

A tetralithium diphosphate obtained by hydrothermal treatment at 1000 atm (1 atm = 101 325 Pa) for 100 h with four formula units of $\text{Li}_4\text{P}_2\text{O}_7$ in a $P2_1/n$ cell of dimensions a = 5.190 (2), b = 13.902 (3), c = 7.901 (3) Å, $\beta = 89.97$ (3)° was described recently (Yakubovich & Mel'nikov, 1994). The structure was described as the first lithium phosphate structure containing diphosphate groups formed by vertex-sharing PO₄ tetrahedra (see also Wenger & Armbruster, 1991; Tien & Hummel, 1961; Keffer, Mighell, Mauer, Swanson & Block, 1967; Averbuch-Pouchot & Durif, 1986). Comparison of the present structure with that described by Yakubovich & Mel'nikov (1994) is not possible as the published coordinates of the latter appear to contain errors.

The new phase reported here (Fig. 1) can be described in terms of P_2O_7 and LiO_4 units. The pyrophosphate groups (lower right-hand side of Fig. 1), built up from two PO₄ tetrahedra sharing a vertex, show typical geometry with two long P—O bonds [P1—O1 1.622 (1) and P2—O1 1.626 (1) Å] corresponding to the P—O— P bridge and three shorter P—O bonds in the range 1.504 (1) to 1.527 (1) Å (Leclaire, Benmoussa, Borel, Grandin & Raveau, 1988; Constentin, Borel, Grandin, Leclaire & Raveau, 1990). The LiO₄ tetrahedra are highly distorted with Li—O distances ranging from



Fig. 1. Projection of the structure on the *ab* plane showing the atom labels. Displacement ellipsoids are represented at the 50% probability level. Symmetry codes: (i) 1 - x, 1 - y, 1 - z; (ii) 2 - x, 1 - y, 2 - z; (iv) 2 - x, 1 - y, 1 - z; (vii) 1 - x, -y, 1 - z; (viii) x, 1 + y, z; (ix) x, 1 + y, 1 + z; (xii) x, y, 1 + z.



Fig. 2. The structure as drawn along the *b* axis of the pseudo monoclinic *A*-centred cell. All possible twofold axes appearing along **b** are destroyed by the relative situation of the P2—O5 and Li2—O6 bonds around them. See, for instance, the pseudo-twofold axis perpendicular to the eight-membered rings formed by the labelled atoms.

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